



# Journées MOSAIC September 25<sup>th</sup> 2024

PhD student – Antoine Dartois (2022 – 2025)  
**In-situ quantitative characterization of  
microstructures in iron-based alloys from binary  
Fe<sub>60</sub>Ni<sub>40</sub> to HEA – Fe<sub>37</sub>Ni<sub>34</sub>Cr<sub>16</sub>Mn<sub>13</sub>**

Supervisors : Estelle Meslin<sup>1</sup>, Maylise Nastar<sup>1</sup>,  
Brigitte Décamps<sup>2</sup>, in collaboration with Anna  
Fraczkiewicz<sup>3</sup>

<sup>1</sup>: CEA/ISAS/DRMP/S2CM/SRMP

<sup>2</sup> IJClab/IN2P3

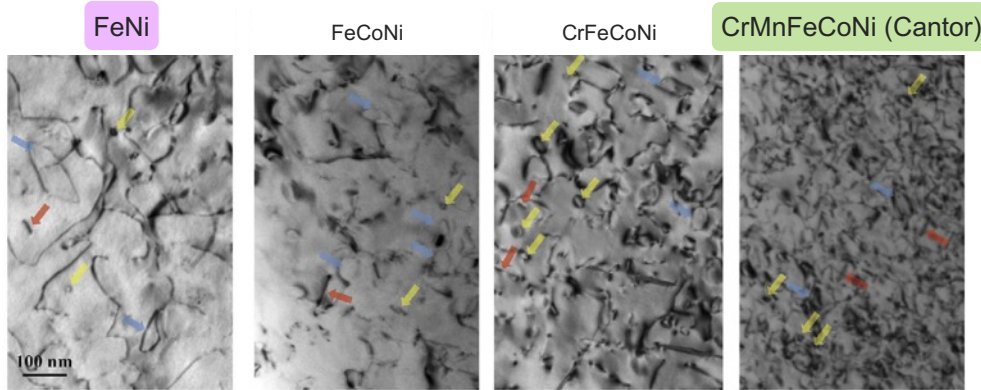
<sup>3</sup> EMSE



# Controversial results on HEA microstructure under irradiation

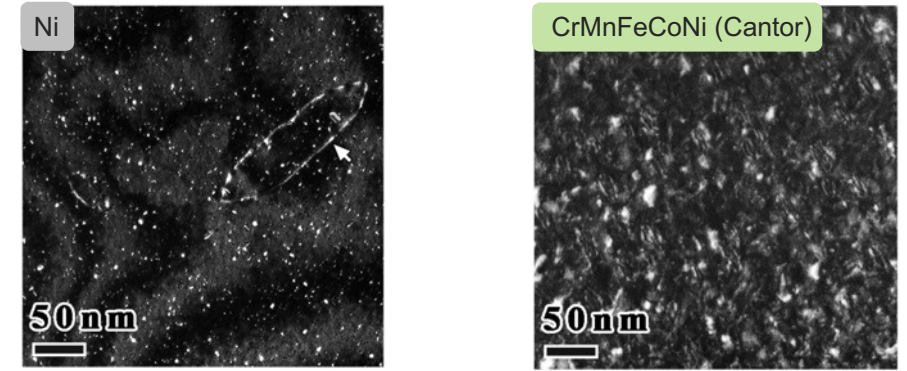
Lu et al, Acta mat (2017)

3 MeV Ni at 773 K up to 38 dpa



Shi et al, JNM (2018)

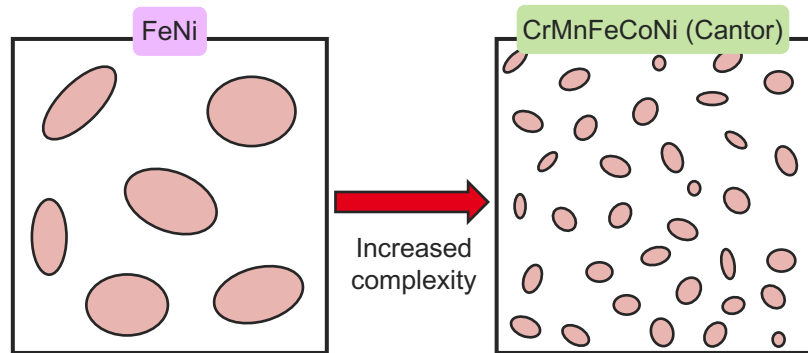
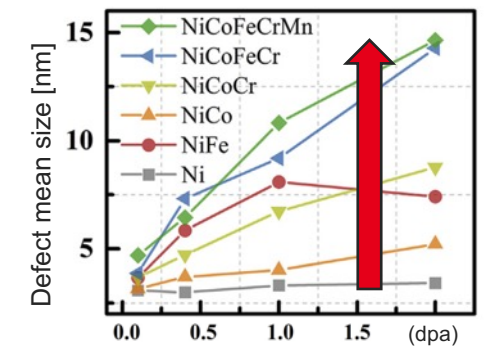
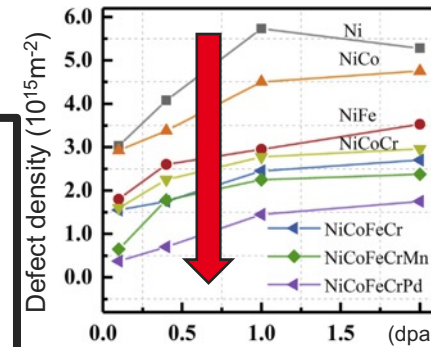
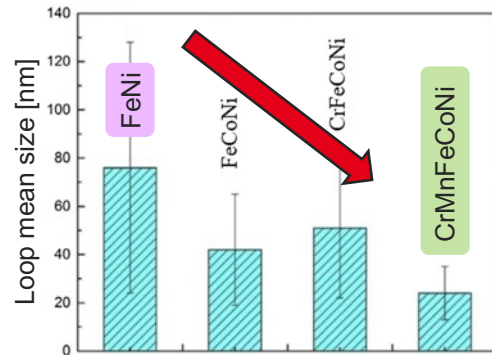
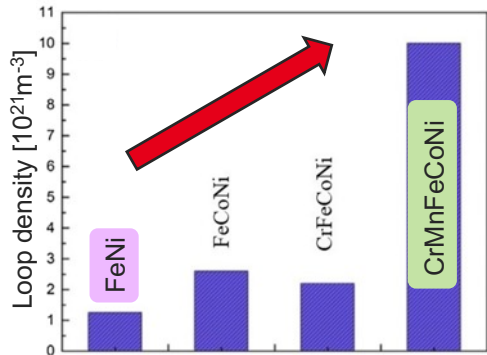
1 MeV Kr at 773 K up to 2 dpa



**✗ Discrepancy**  
High impact of irradiation conditions and dose on microstructure

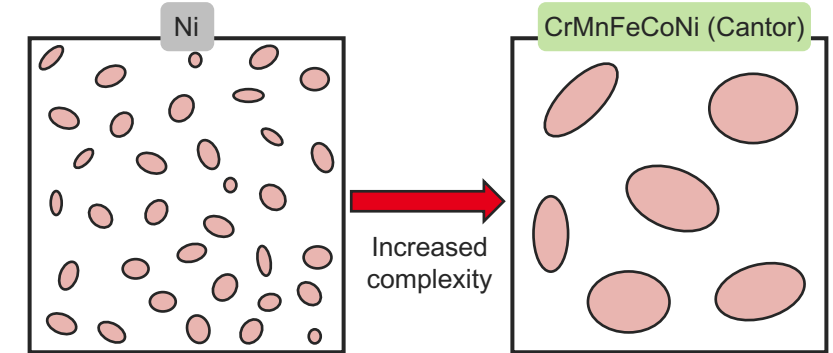
**? Questions**

- ❖ What about HEAs without Co?
- ❖ How differently do HEAs and stainless steels behave?



Low density of large loops

Higher density of smaller loops



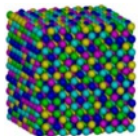
High density of small loops

Lower density of larger loops

# Experimental details

3 compositions

ES1



1

Model high-entropy alloy  
 $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$

2

Industrial austenitic steel  
**316L**

$\text{Fe}_{\sim 62}\text{Ni}_{\sim 12}\text{Cr}_{\sim 17}$   
& (Mo, Mn, Si)

3

Model binary alloy  
 $\text{Fe}_{60}\text{Ni}_{40}$

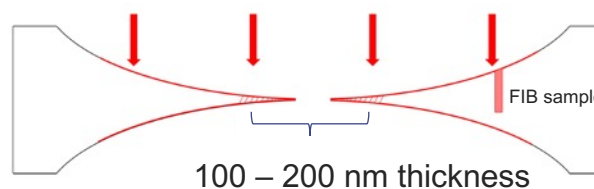
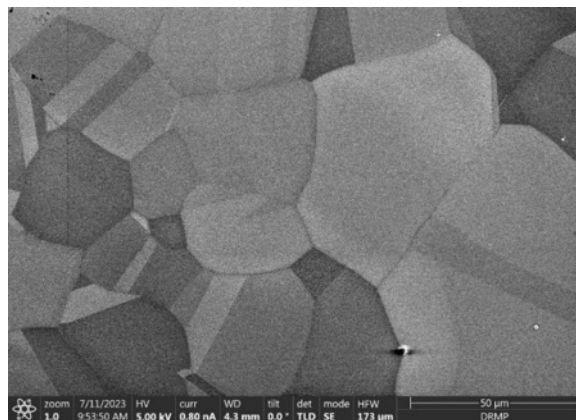
## Alloy fabrication

- High purity raw materials (< 20 ppm of C, N, O, S)
- Cold crucible melting
- Hot forging (about 1000 °C), thermomechanical treatment homogenization (1000 °C / 2h)

## Sample preparation

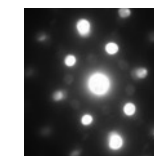
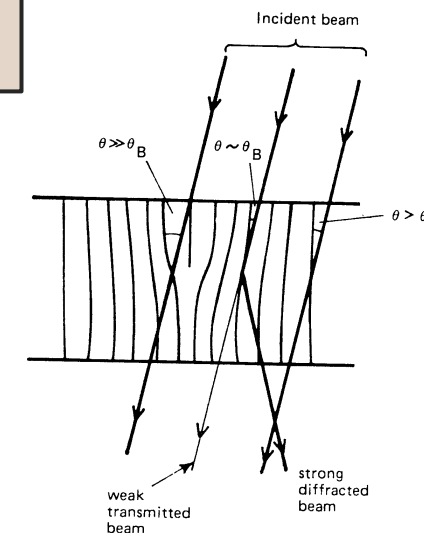
Thin foils for irradiation:

- Mechanical polishing
- Thermal annealing
- Electro-polishing
- Focused ion beam



## Conventional TEM – defect analysis

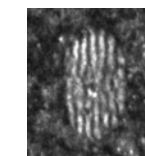
Bragg law:  
 $n\lambda = 2d \sin \theta_B$



Diffraction pattern

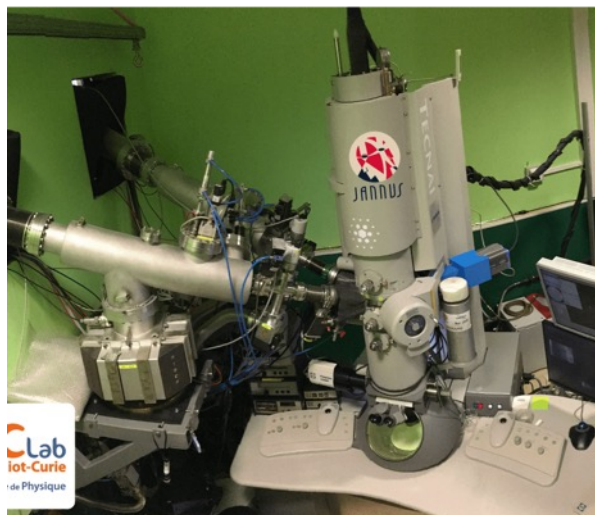


Bright field (BF)



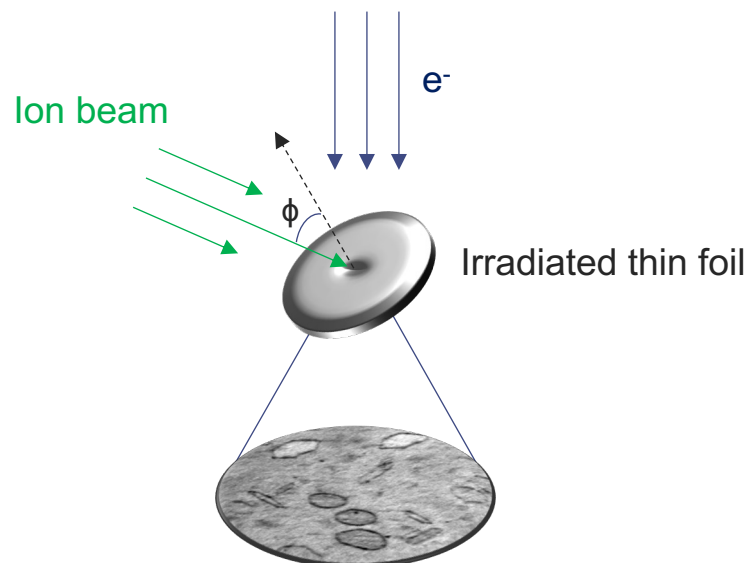
Weak-beam dark field (WBDF)

# Model in-situ ion irradiations to simulate neutron irradiation

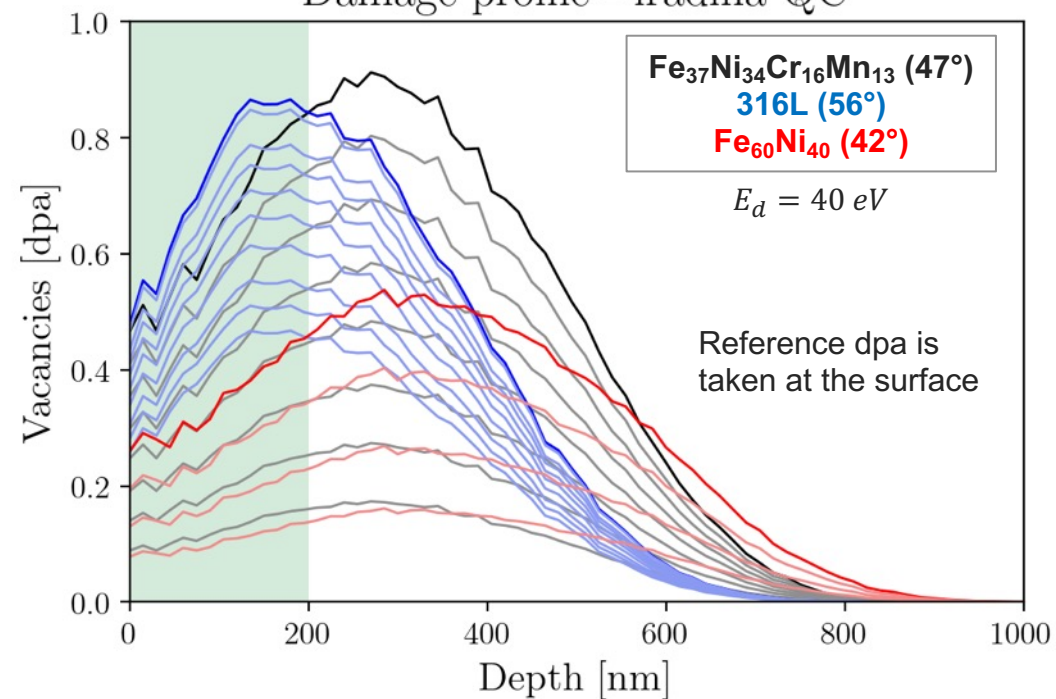


Temperature  
550 °C

Material	Ion	G [dpa/s]	Ion beam incidence angle ( $\phi$ )
$\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$	$\text{Fe}^+$ 2 MeV	$1,9\text{E}-4$	$47^\circ$
<b>316L</b>	$\text{Fe}^+$ 2 MeV	$1,5\text{E}-4$	$56^\circ$
$\text{Fe}_{60}\text{Ni}_{40}$	$\text{Ni}^+$ 2 MeV	$2,2\text{E}-4$	$42^\circ$



Damage profile - irradiation QC

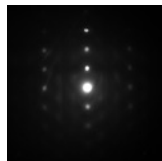


# Microstructure evolution under irradiation

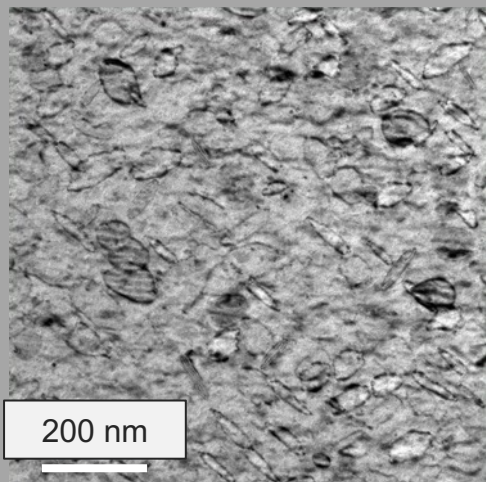
**Fe<sub>37</sub>Ni<sub>34</sub>Cr<sub>16</sub>Mn<sub>13</sub>**

→ **0.44 dpa**

Z = [012]  
g = [200]



Loop size: ++  
Density: +  
Growth: +



Conventional TEM / BF

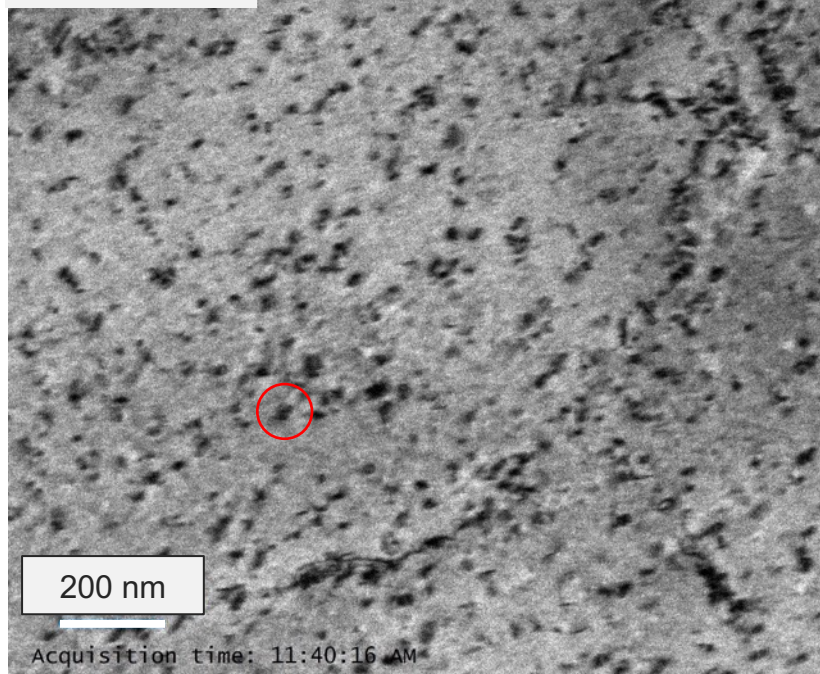
**316L**

→ **0.54 dpa**

Z = [341]  
g = [1-11]



Loop size: -  
Density: -  
Growth: -

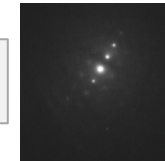


Conventional TEM / BF

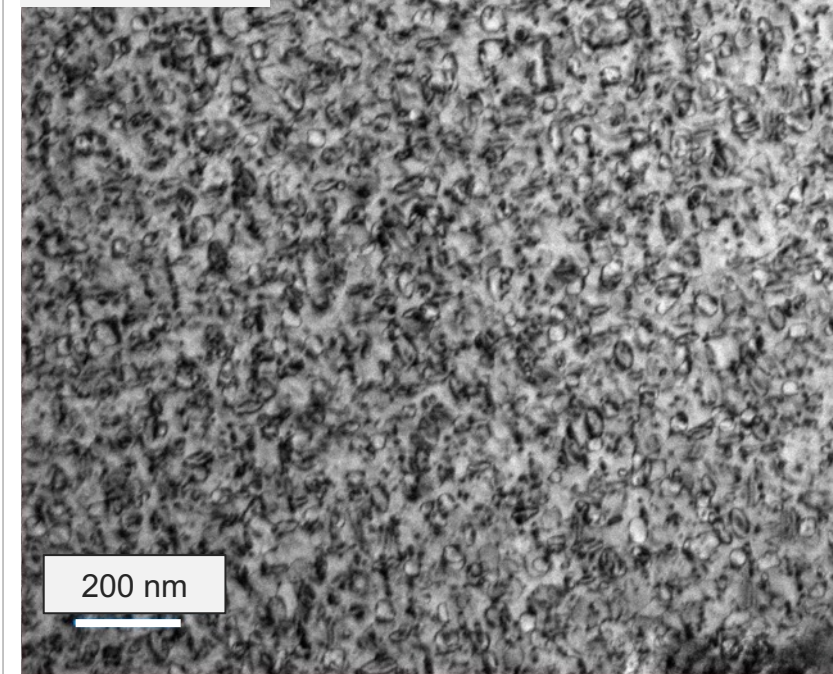
**Fe<sub>60</sub>Ni<sub>40</sub>**

→ **0.28 dpa**

Z = [001]  
g = [200]



Loop size: +  
Density: ++  
Growth: ++



Conventional TEM / BF

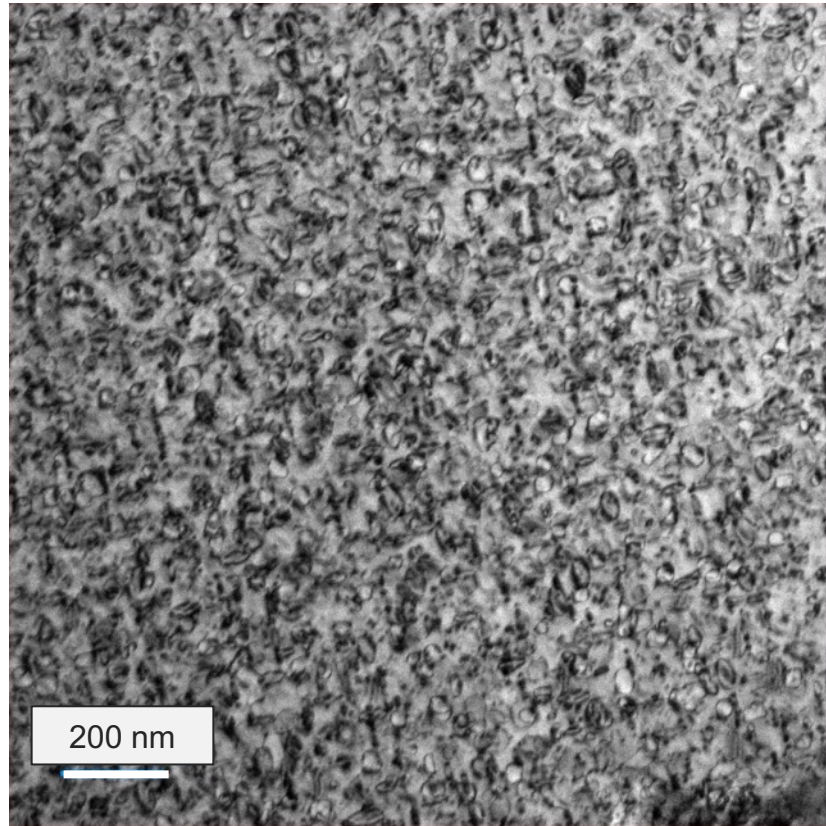
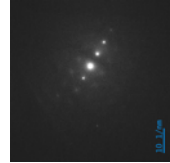
# Method – Loop size distribution

example for Fe<sub>60</sub>Ni<sub>40</sub>

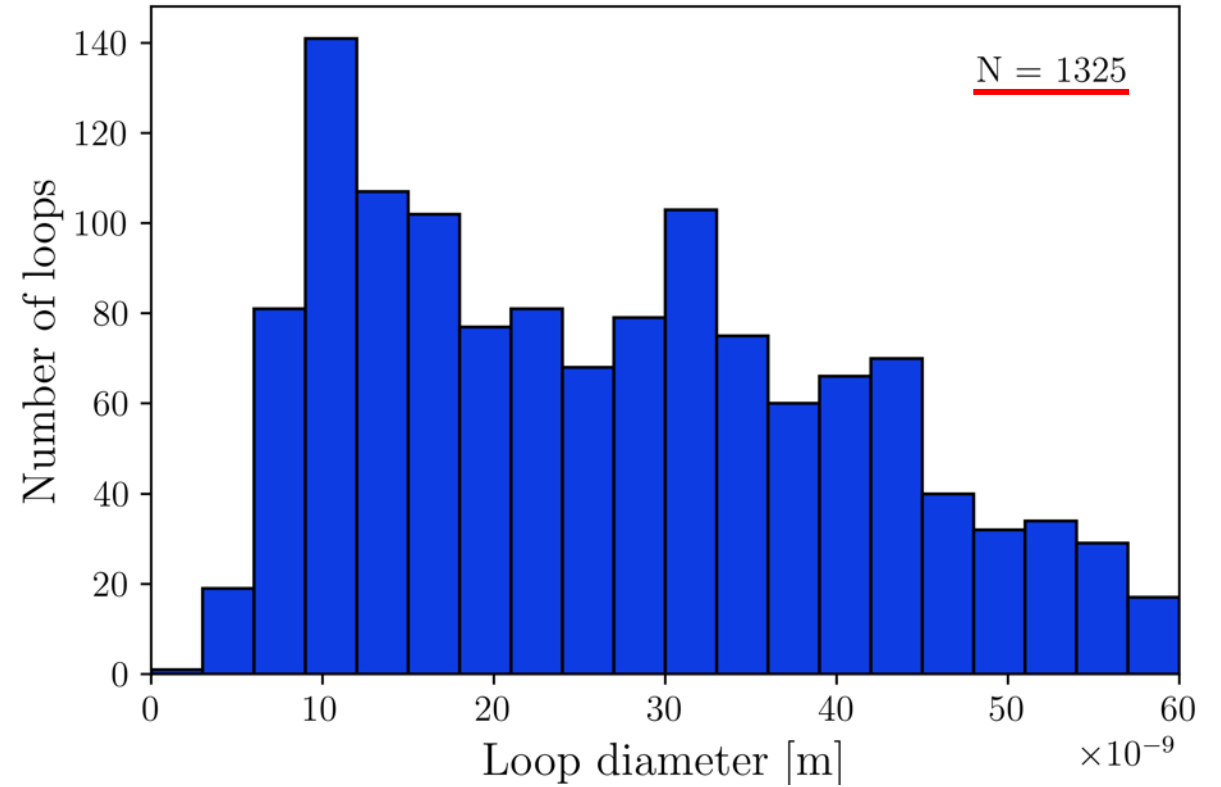


0.28 dpa

Z = [001]  
g = [200]

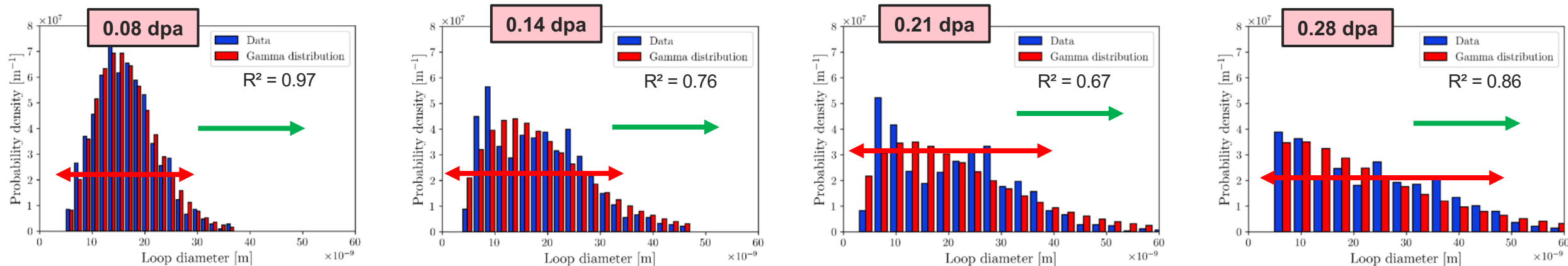


Conventional TEM / BF



# Method – Gamma distribution fit

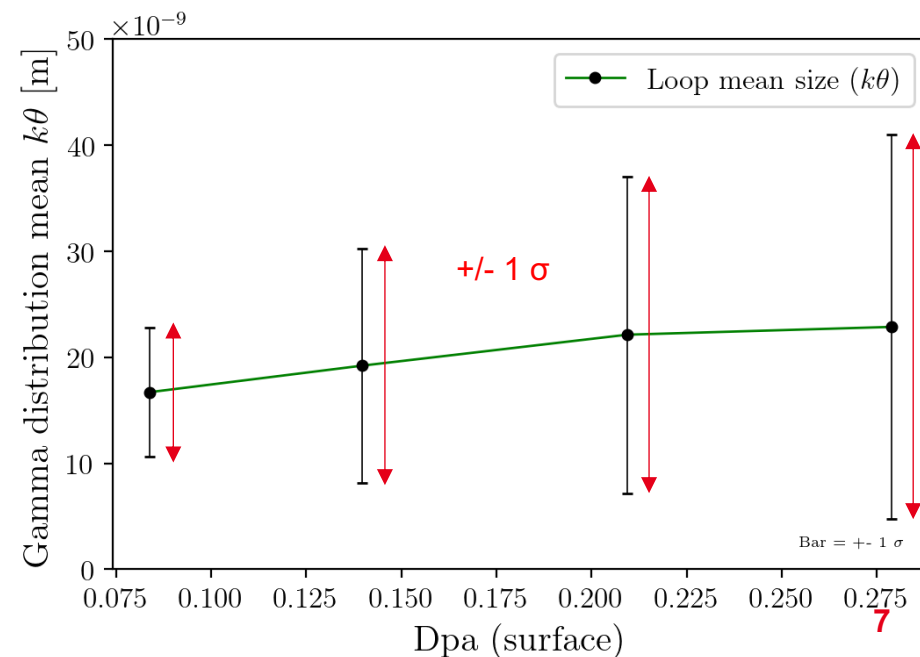
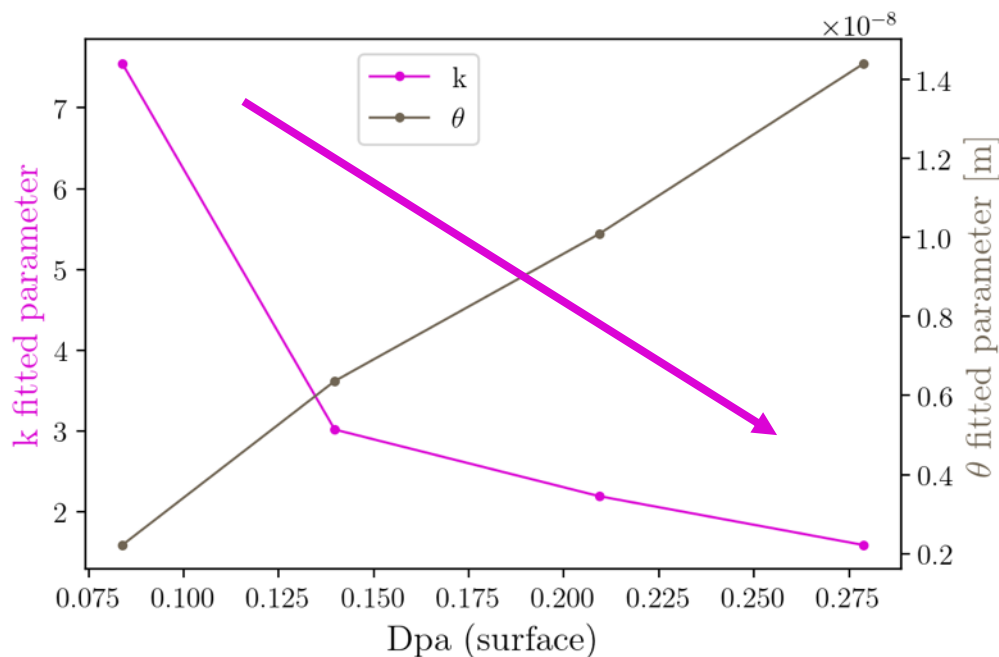
example for Fe<sub>60</sub>Ni<sub>40</sub>



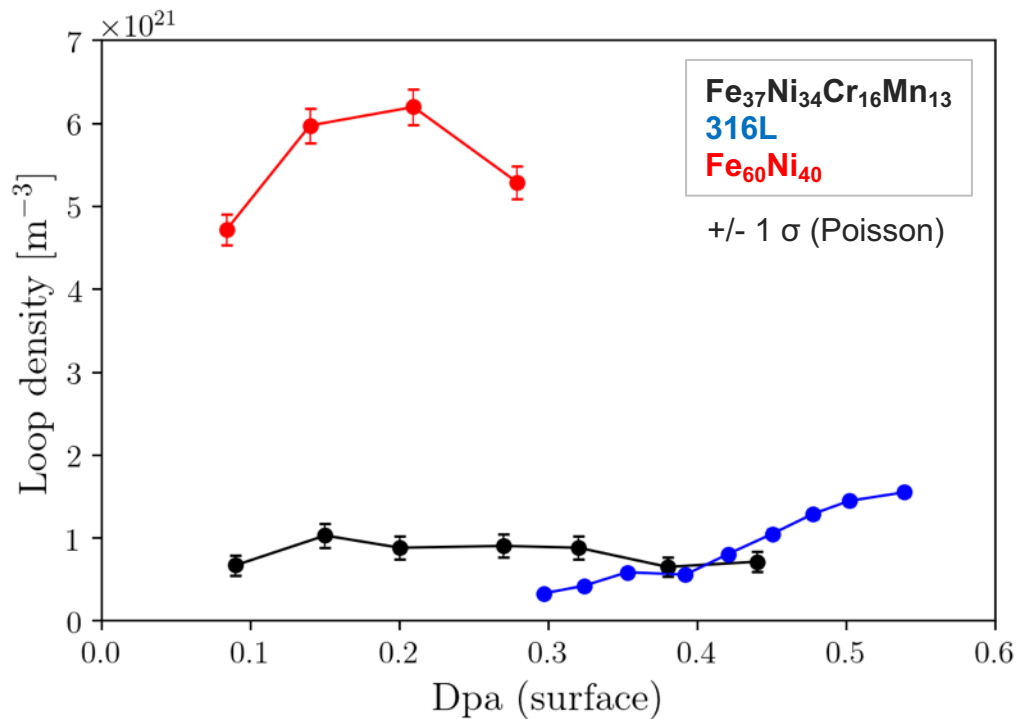
As damage increases...

❖ Distribution **shift** → and **spreading** ↔

**Gamma distribution**  
 $X \sim \Gamma(k, \theta)$   
 $f(x; k, \theta) = \frac{x^{k-1} e^{-x/\theta}}{\Gamma(k)\theta^k}$   
*k*: shape parameter  
 $\theta$ : scale parameter  
 $\mu = k\theta$   
 $\sigma^2 = k\theta^2$

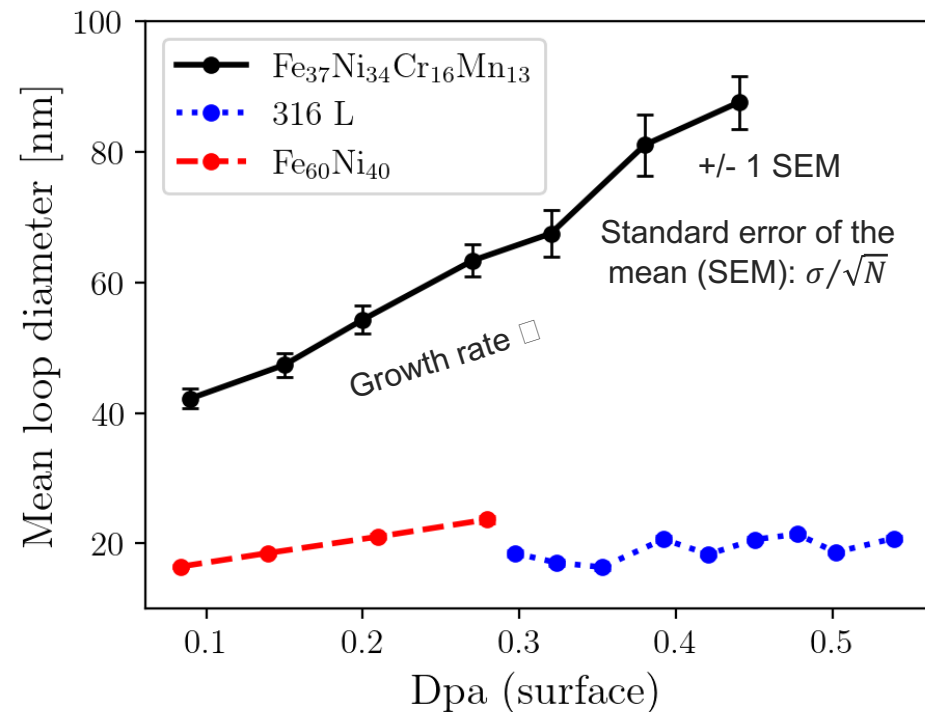


# In-situ loop nucleation and growth rates



**1** Loop nucleation rates

**Fe<sub>60</sub>Ni<sub>40</sub> > Fe<sub>37</sub>Ni<sub>34</sub>Cr<sub>16</sub>Mn<sub>13</sub> > 316L**



	Fe <sub>37</sub> Ni <sub>34</sub> Cr <sub>16</sub> Mn <sub>13</sub>	316L	Fe <sub>60</sub> Ni <sub>40</sub>
a [m/dpa]	1,33E-7	1,38E-8	3,67E-8
b [m]	2,82E-8	1,34E-8	1,34E-8
R <sup>2</sup>	0,99	0,40	1,00

**2** Loop growth rates

**Fe<sub>37</sub>Ni<sub>34</sub>Cr<sub>16</sub>Mn<sub>13</sub> > Fe<sub>60</sub>Ni<sub>40</sub> > 316L**

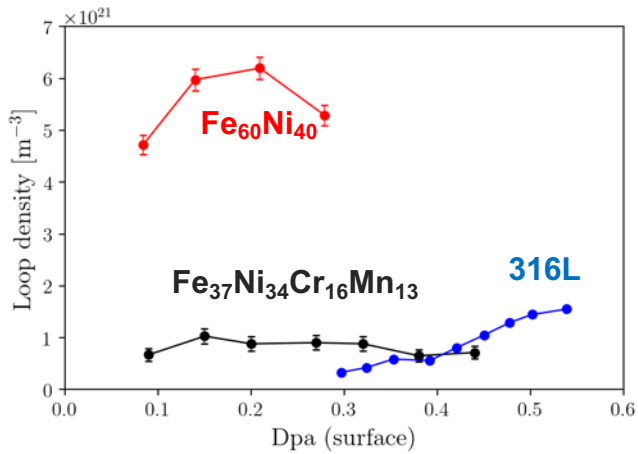


# In-situ loop nucleation and growth rates

1

## Loop nucleation rates

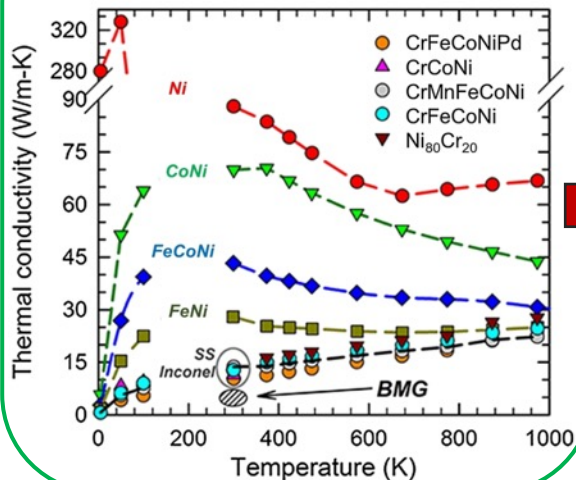
$\text{Fe}_{60}\text{Ni}_{40} > \text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13} > \text{316L}$



### Possible interpretation

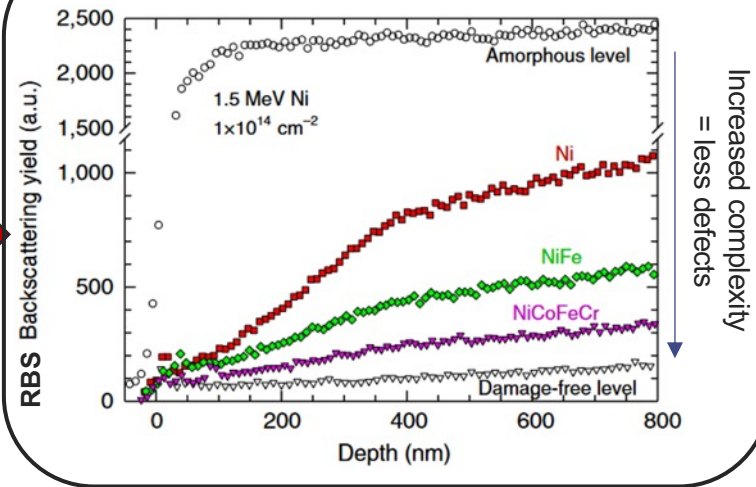
Cluster production within the displacement cascade is reduced in  $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$  and **316L**.  
A self-recovery tendency due to decreased thermal conductivity?

Slower energy dissipation  
⇒ more recombination



Jin et al. (2018)

### Reduced damage



Zhang et al, Nature Com (2015)

2

## Loop growth rates

$\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13} > \text{Fe}_{60}\text{Ni}_{40} > \text{316L}$

Material	Distribution growth rate (nm/dpa)	Individual growth rate (nm/dpa)
$\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$	$133 \pm 7$	$154 \pm 17$
316 L	$14 \pm 6$	$68 \pm 8$
$\text{Fe}_{60}\text{Ni}_{40}$	$37 \pm 0.3$	$97 \pm 13$

Kiritani (1994)

$$\frac{dr}{dt} = \frac{2}{b} (Z_i - Z_l) \sqrt{\frac{GV_{at}D_{ol}}{R_{il}}} \exp\left(-\frac{E_v^m}{2k_bT}\right)$$

$$\frac{dr}{dt} \propto \exp\left(-\frac{E_v^m}{2k_bT}\right)$$

### Possible interpretation

Enhanced migration of vacancies in  $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$   
Lower  $E_v^m$  ?

# Estimation of $E_m^v$ in 316L – Kiritani model

Kiritani (1994)

$$\frac{dr}{dt} = \frac{2}{b} (Z_i - Z_l) \sqrt{\frac{GV_{at} D_{ol}}{R_{il}}} \exp\left(-\frac{E_v^m}{2k_b T}\right)$$

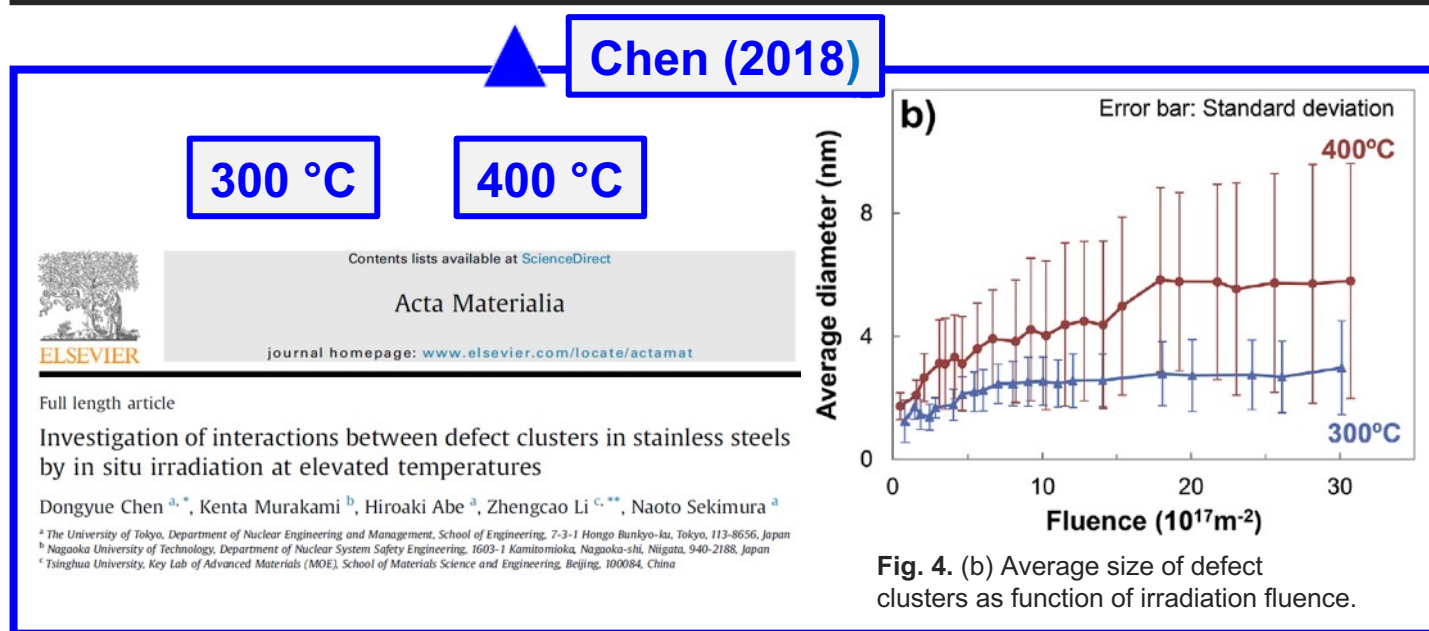
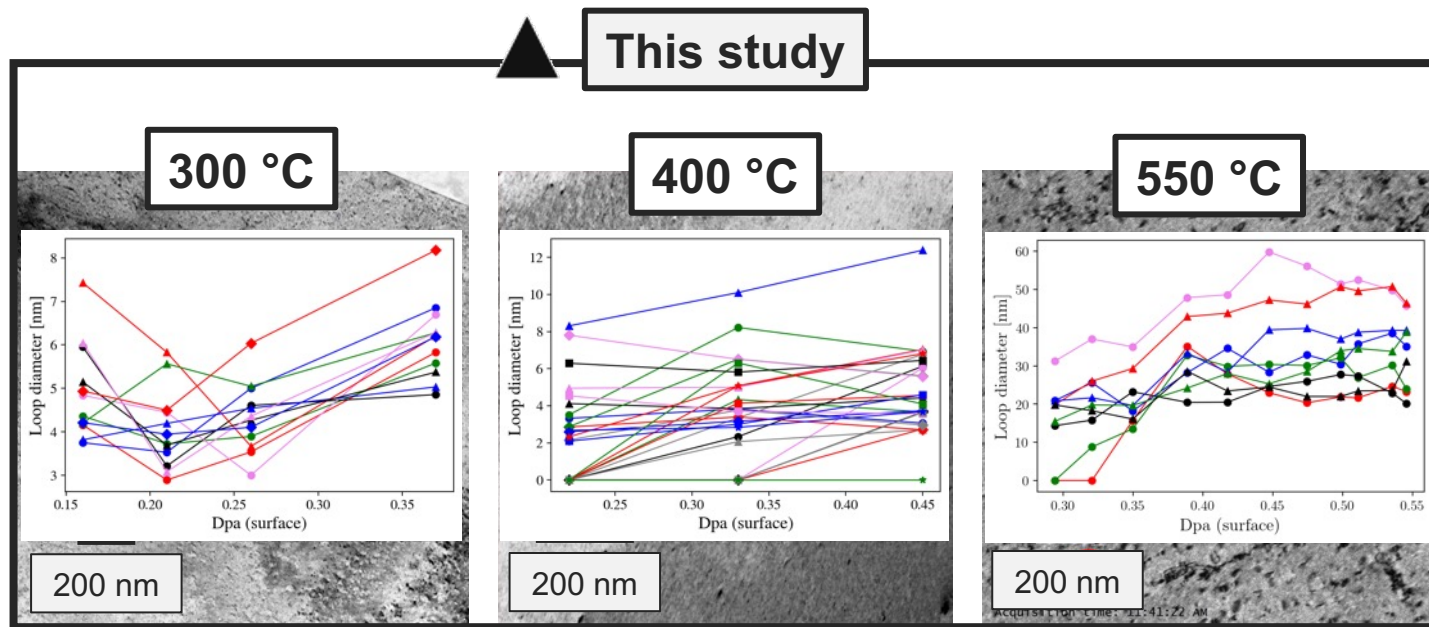
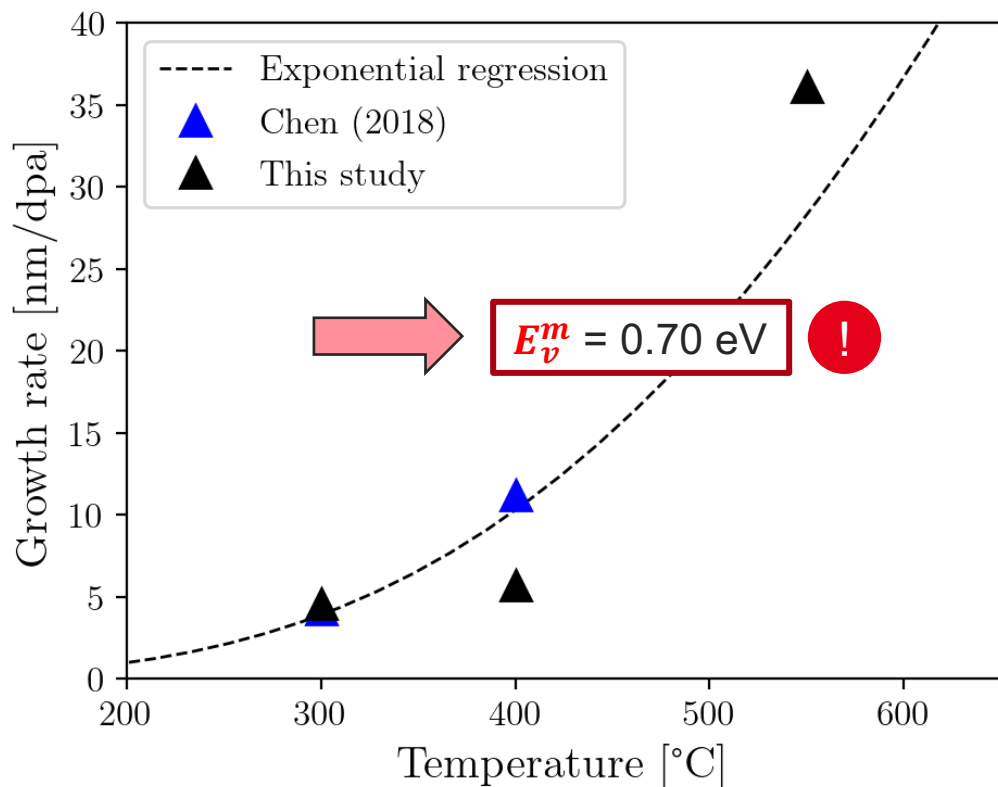
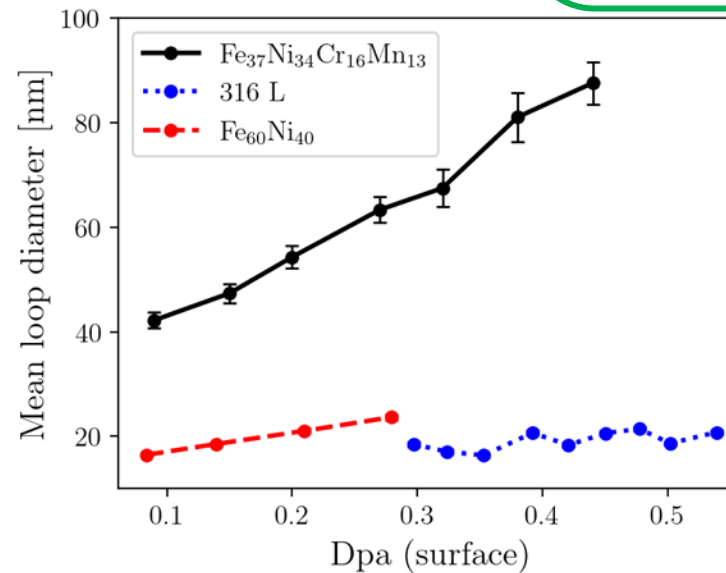
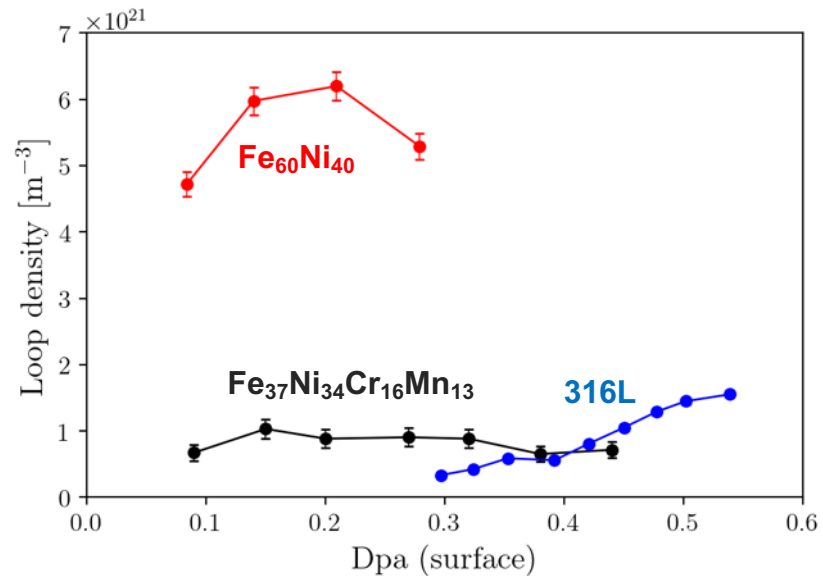
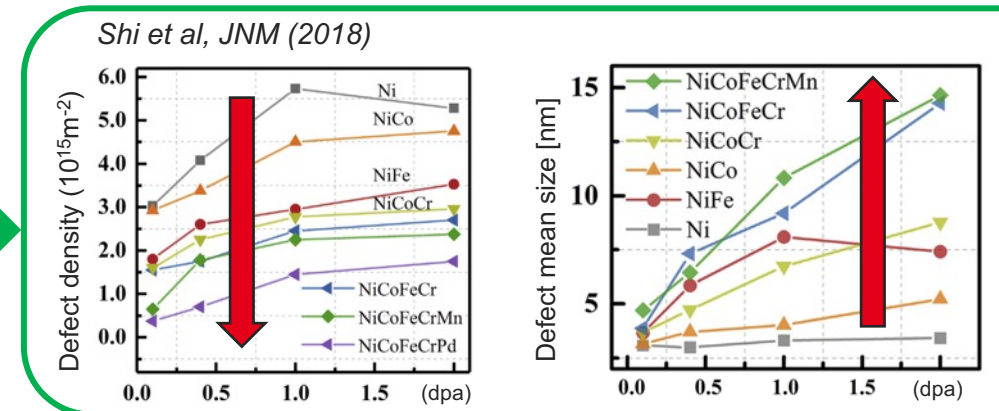
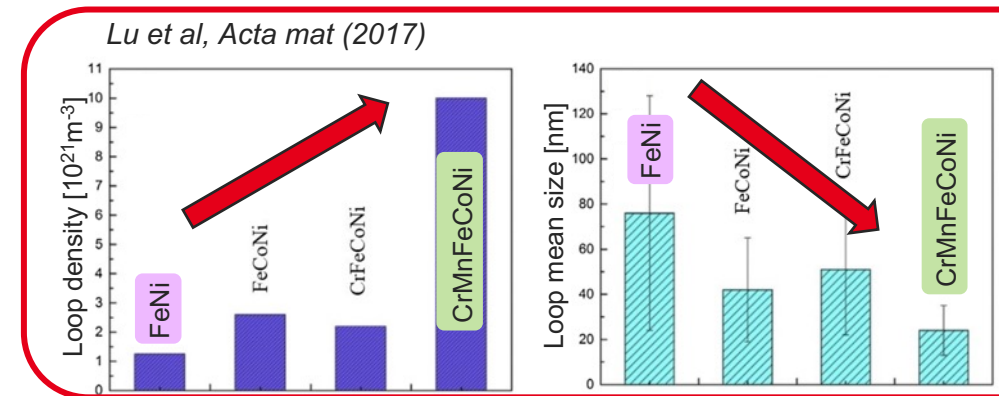


Fig. 4. (b) Average size of defect clusters as function of irradiation fluence.

# Conclusions

- Quantitative and statistical study of  $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$ ,  $\text{Fe}_{60}\text{Ni}_{40}$  and 316L, in-situ irradiated using Ni/Fe 2 MeV ions at 550°C
- Significant differences in **damage kinetics** between  $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13}$ ,  $\text{Fe}_{60}\text{Ni}_{40}$  and 316L:
  - Loop density:  $\text{Fe}_{60}\text{Ni}_{40} > \text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13} > 316\text{L}$ 
    - One interpretation: Thermal conductivity affecting the cluster formation within the cascade?
  - Loop growth rate:  $\text{Fe}_{37}\text{Ni}_{34}\text{Cr}_{16}\text{Mn}_{13} > \text{Fe}_{60}\text{Ni}_{40} > 316\text{L}$ 
    - One interpretation: Faster migration of vacancies?





# Thank you for your attention!



Special thanks to JANNuS Orsay team